

Supporting Information

La_{0.2}Sr_{0.25}Ca_{0.45}TiO₃ surface reactivity with H₂: a combined *operando* NEXAFS and computational study

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XRD Patterns

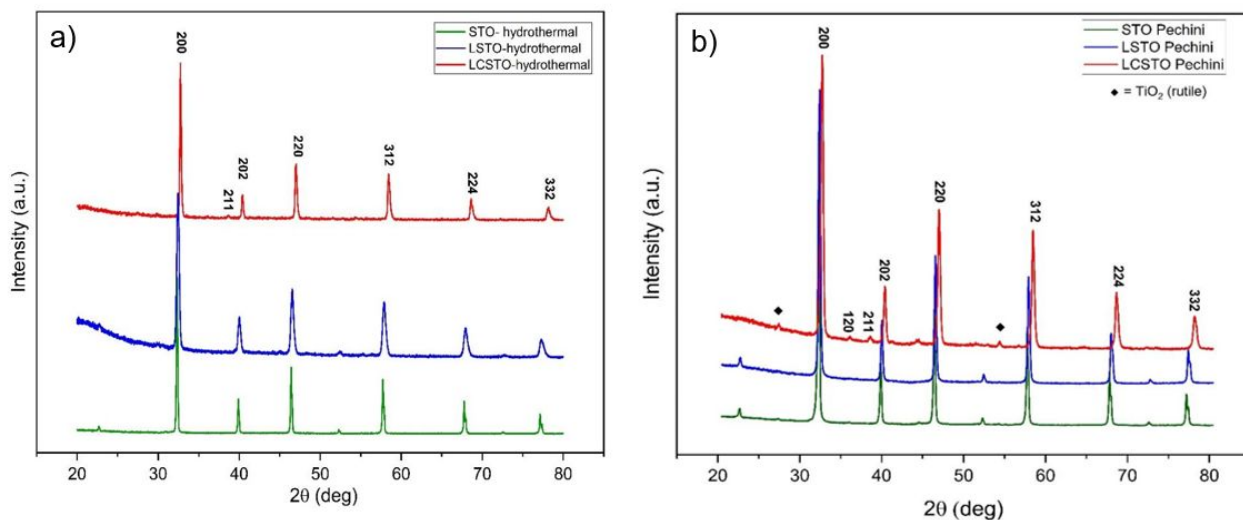


Fig. S1 XRD pattern of the samples synthesized by hydrothermal route (a) and modified-Pechini (b)

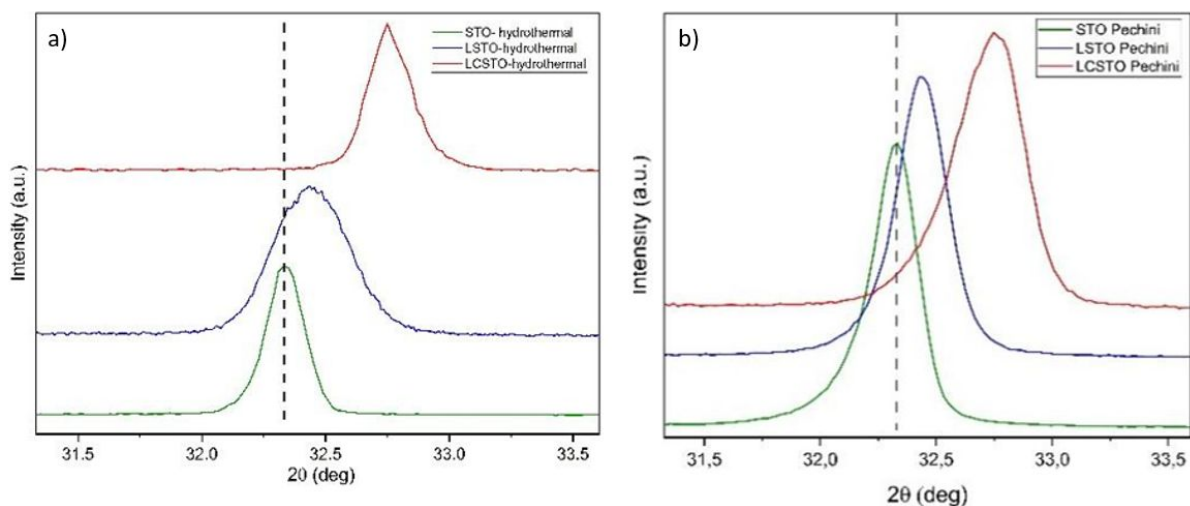


Fig. S2 Inset on the (200) reflection of figure S1

	LCSTO Hydrothermal	LCSTO Pechini
a (\AA)	5.5419	5.5432
b (\AA)	5.4974	5.4937
c (\AA)	7.7995	7.7966

Table S1: LCSTO cell parameters calculated from XRD

Particle size analysis

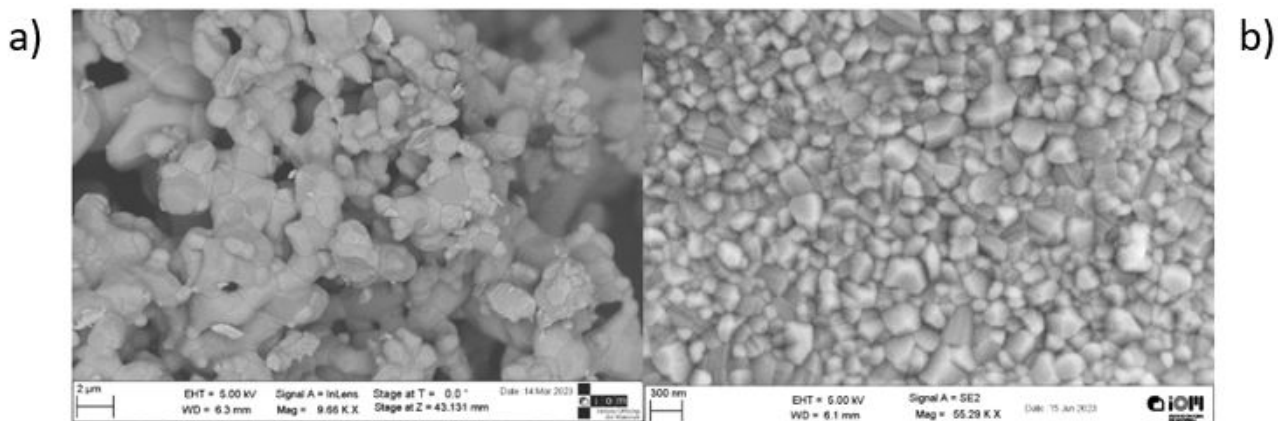


Fig. S3 SEM images of LCSTO Hydrothermal (a) and LCSTO Pechini (b)

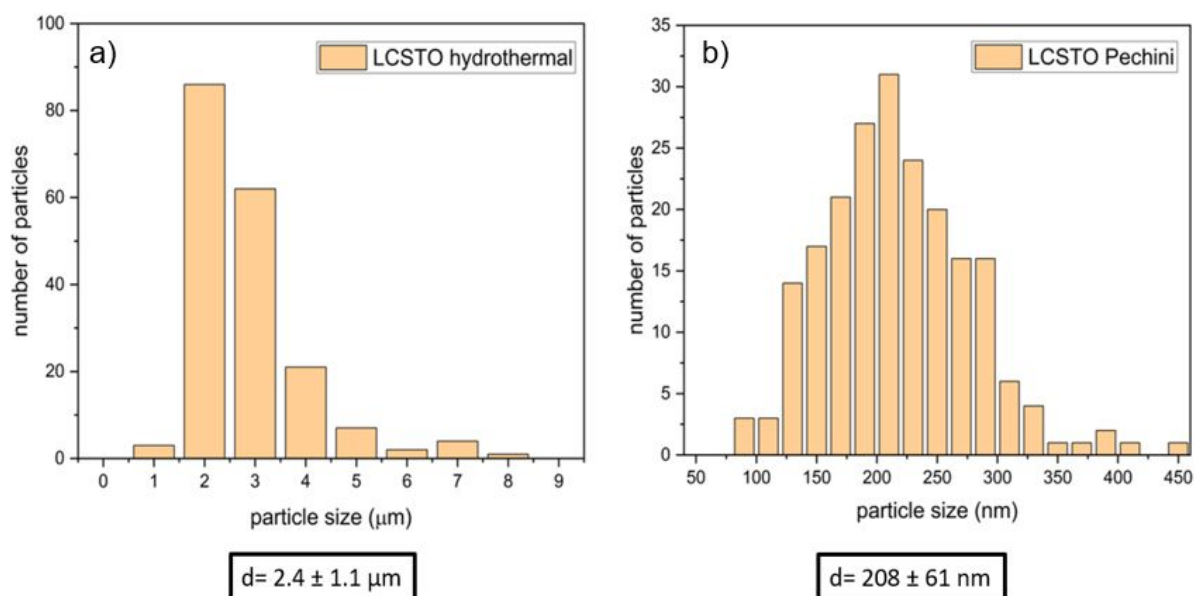


Fig. S4 Size distribution of LCSTO hydrothermal (a) and LCSTO Pechini (b)

XPS Surveys

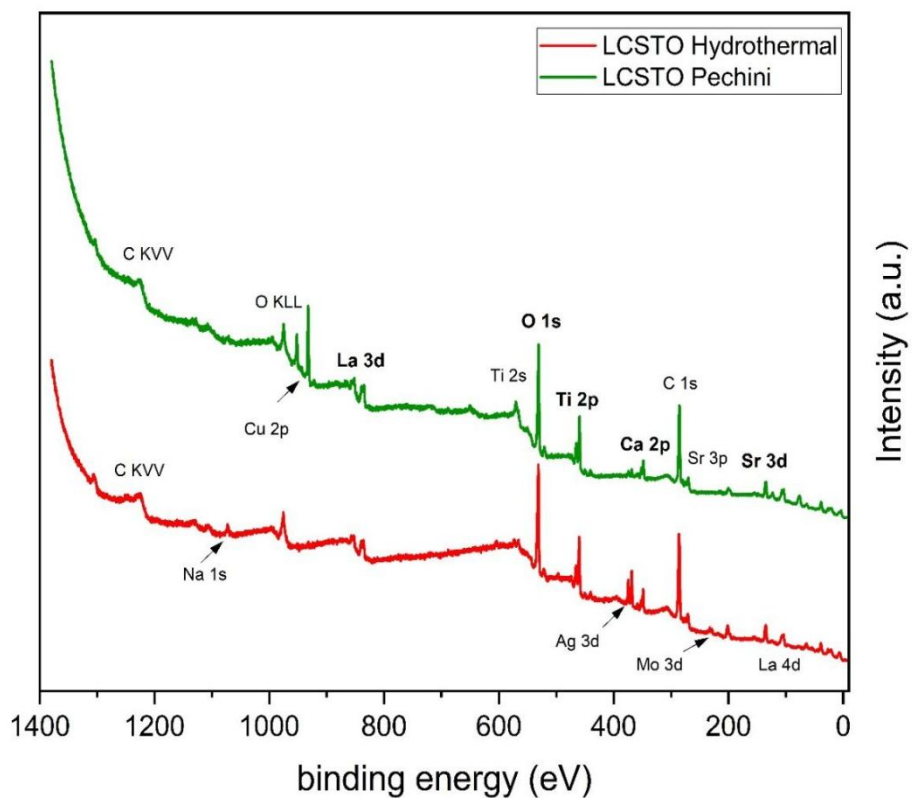
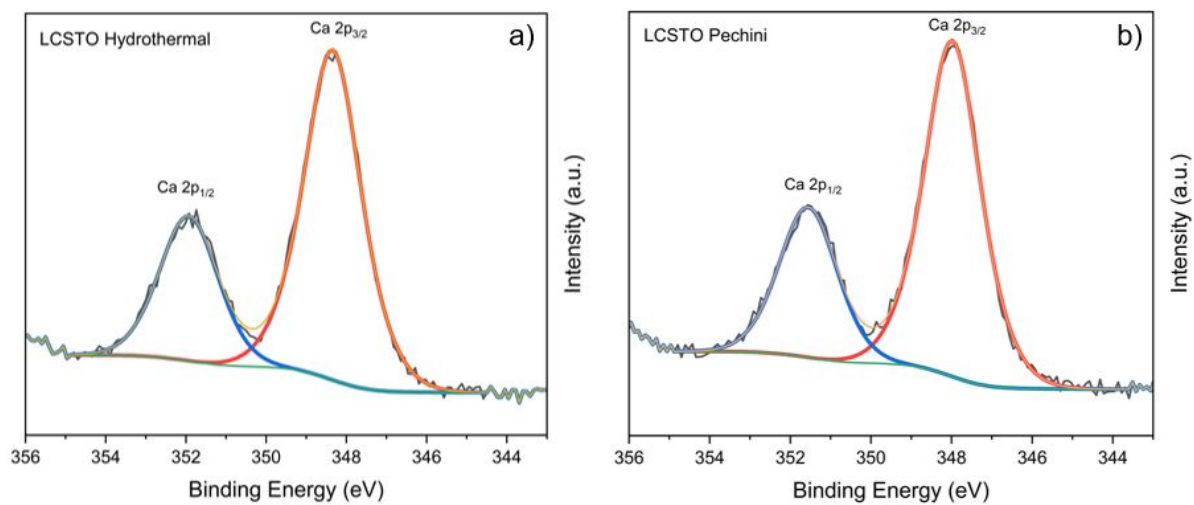
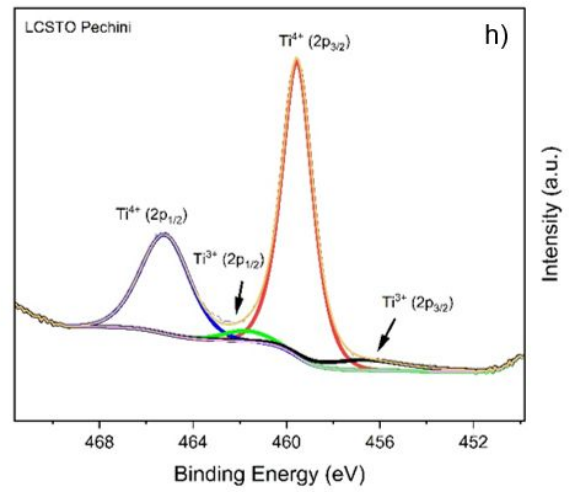
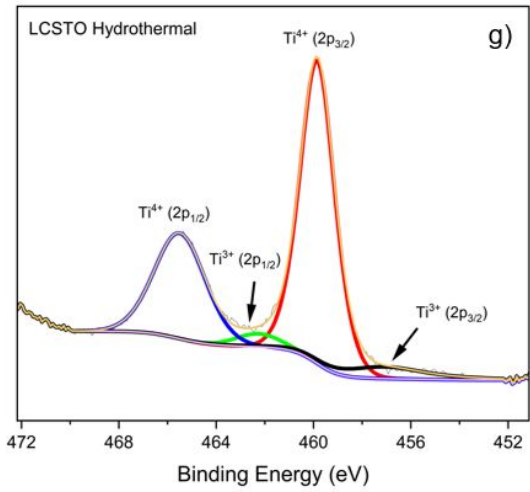
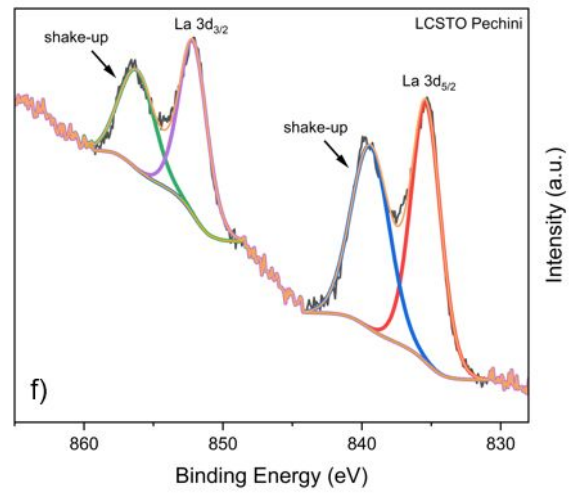
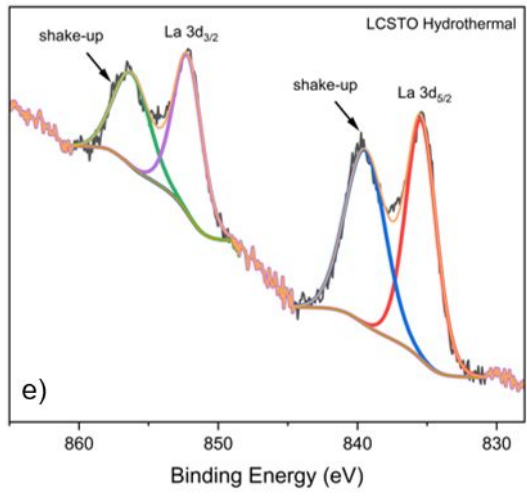
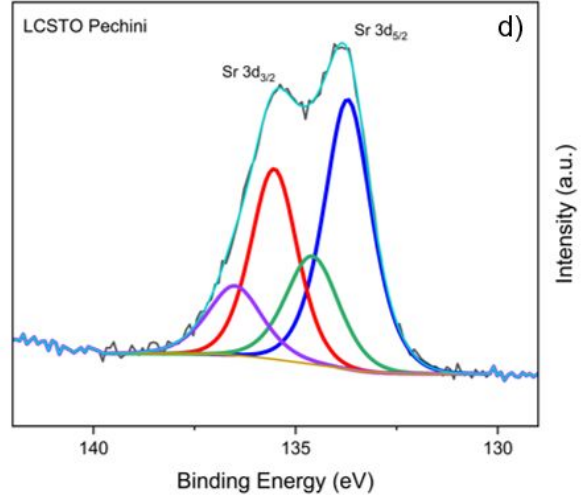
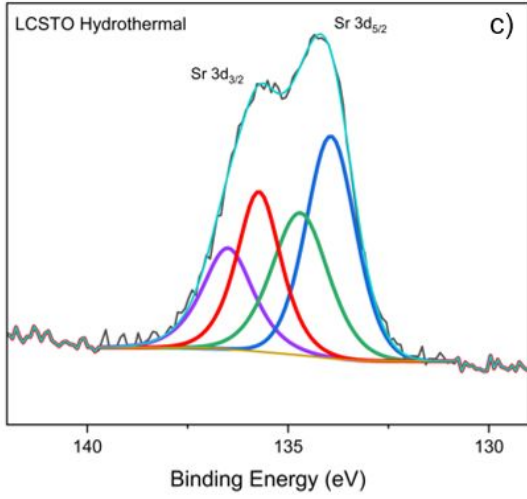


Fig. S5 Surveys of LCSTO samples (Cu and Mo peaks are ascribed to the sample holders, while Ag to silver paint)

XPS core levels fittings





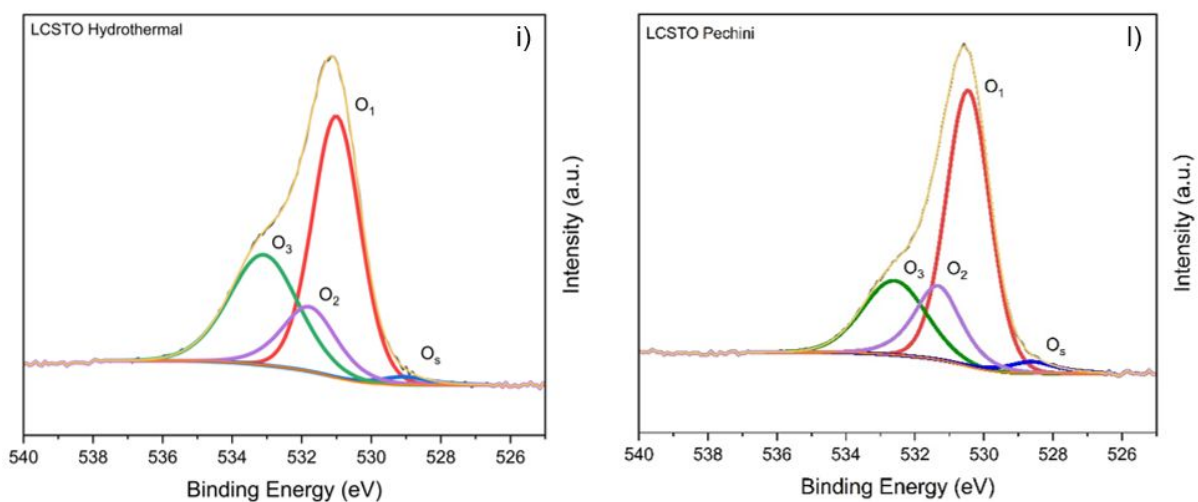


Fig. S6 XPS core levels fittings for LCSTO Hydrothermal and Pechini

O1s LCSTO Hydrothermal					
	Position	FWHM	Line shape	Area	relative %
Lattice (O ₁)	531.0 eV	1.61 eV	GL (30)	92521.3	51.34
Vacancies (O ₂)	531.78 eV	1.86 eV	GL (60)	27610.7	15.32
-OH (O ₃)	533.09 eV	2.33 eV	GL (40)	56990.3	31.62
Sample holder (O _s)	529.06 eV	1.61 eV	GL (80)	3103.9	1.72
Residual STD= 1.048					

Table S2: XPS fitting parameters for O1s: sample LCSTO Hydrothermal

O1s LCSTO Pechini					
	Position	FWHM	Line shape	Area	relative %
Lattice (O ₁)	530.46 eV	1.48 eV	GL (30)	74316.6	57.20
Vacancies (O ₂)	531.33 eV	1.59 eV	GL (70)	22813.1	17.56
-OH (O ₃)	532.59 eV	2.15 eV	GL (30)	29074	22.38
Sample holder (O _s)	528.61 eV	1.64 eV	GL (80)	3711.9	2.86
Residual STD= 1.238					

Table S3: XPS fitting parameters for O1s: sample LCSTO Pechini

SAMPLE	Theoretical Stoichiometry	XPS outcome
LCSTO-Hydrothermal	$\text{La}_{0.2}\text{Sr}_{0.25}\text{Ca}_{0.45}\text{TiO}_3$	$\text{La}_{0.20}\text{Sr}_{0.40}\text{Ca}_{0.50}\text{TiO}_3$
LCSTO-Pechini	"	$\text{La}_{0.20}\text{Sr}_{0.40}\text{Ca}_{0.50}\text{TiO}_3$
LCSTO-MBE pristine	"	$\text{La}_{0.15}\text{Sr}_{0.40}\text{Ca}_{0.40}\text{TiO}_3$
LCSTO-MBE "spent"	"	$\text{La}_{0.30}\text{Sr}_{0.70}\text{Ca}_{0.50}\text{TiO}_3$

Table S4: LCSTO estimated stoichiometry by XPS

Experimental procedure for the *Operando* NEXAFS experiments

All the samples were subjected to the same experimental procedure: firstly, the sample was pre-heated to 120 °C in pure He flow to remove the adsorbed water and possible contaminants. Then, it was cooled down to RT and heated up to 350 or 400 °C in H₂ (100%). A graphical scheme of the experimental procedure is reported in fig. S7 for clarity.

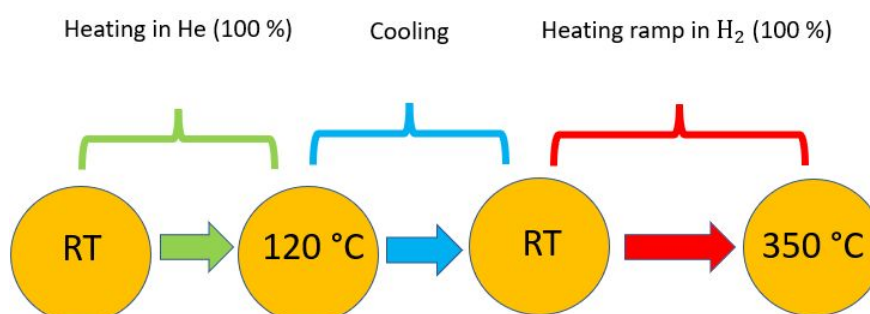


Fig. S7 Experimental scheme of the measurements

Detailed description of STO, LSTO, LCSTO NEXAFS features

Starting the description from lower photon energies, L₃ edge shows two pre-edge peaks (**a** and **b** in Fig. 2)) located at ~ 460.5 and 461.5 eV, assigned to transitions towards virtual orbitals of t_{2g}-like symmetry^{1,2}, originated from 2p⁶ d⁰ → 2p⁵ d¹ transition for Ti⁴⁺ in O_h symmetry and characteristic of d⁰ compounds³. The presence of the crystal field in proximity of the octahedral Ti⁴⁺ sites cause the loss of the 3d orbitals degeneracy, giving

rise to the well-known t_{2g} and e_g states: this is reflected by the splitting of the L edges white line. In detail, **c** and **e** structures can be attributed to $2p^{3/2} \rightarrow 3d_{t_{2g}}$ and $2p^{1/2} \rightarrow 3d_{t_{2g}}$ transitions while **d** and **f** to $2p^{3/2} \rightarrow 3d_{e_g}$ and $2p^{1/2} \rightarrow 3d_{e_g}$. It is important to highlight that, in the presence of O_h symmetry, e_g states are related to d_{z^2} and $d_{x^2-y^2}$ orbitals which are strongly hybridized with the ligands (in the case of $SrTiO_3$ with O atoms) because of their geometry. This results in a major sensitivity of the **d** and **f** structures to the Ti-O bonds modifications in the NEXAFS spectra, that could originate from structural distortions or oxygen vacancy formation. Vibronic effects and a larger dispersion caused by a higher degree of hybridization of the Ti 3d e_g orbitals with the surrounding O atoms also reflects in the broadening of **d** and **f**^{2,3}.

Theoretical Modelling

For $SrTiO_3$, the theoretical spectra have been calculated starting from the corresponding crystallographic structure (*Acta Cryst.* (1995). **B51**, 942-951 <https://doi.org/10.1107/S0108768195003752>, 10.1103/PhysRevB.60.2961). The .cif file containing the crystallographic parameters was employed to build a $SrTiO_3$ cluster (fig. S8) that was used for the theoretical simulations of the spectra.

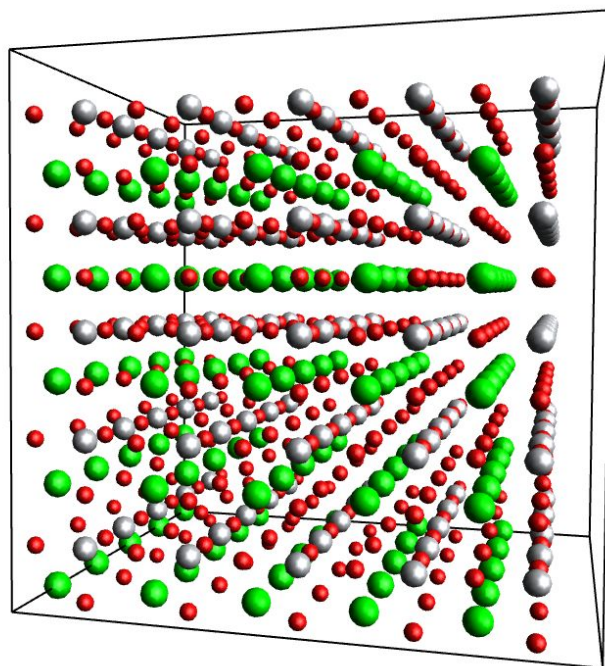


Figure S8. $5 \times 5 \times 5$ $SrTiO_3$ supercell model used for the theoretical simulations. The dimensions of the unit cell are $a=b=c = 3.908 \text{ \AA}$, $\alpha=\beta=\gamma=90^\circ$.

The photoabsorber atom has been chosen in order to be at the centre of the cluster, therefore with crystallographic coordinates $x=0$, $y=0$ and $z=0$. The same procedure has been employed to build analogue clusters with different crystallographic parameters and chemical compositions, described in detail hereafter. In order to build the $\text{La}_{0.2}\text{Sr}_{0.25}\text{Ca}_{0.45}\text{TiO}_3$ cluster, the SrTiO_3 crystal structure was chosen as a starting point. To have a clear view of the operations carried out on the SrTiO_3 structure, a simplified picture of its cubic unit cell is shown in fig. S9, where the first and second coordination shells of the photoabsorber atom are clearly observable (Model 1). In detail, the central Ti site is bound to six oxygen atoms in octahedral coordination, while eight Sr atoms occupy the corners of the resulting FCC cell. In order to build the $\text{La}_{0.2}\text{Sr}_{0.25}\text{Ca}_{0.45}\text{TiO}_3$ model, two operations have been performed on the SrTiO_3 structure:

- Structural modification: $\text{La}_{0.2}\text{Sr}_{0.25}\text{Ca}_{0.45}\text{TiO}_3$ has an orthorhombic unit cell. The cell parameters have been calculated from the XRD patterns (table S1); the values found were then replaced to those of the SrTiO_3 structure when building the cluster model. The resulting orthorhombic unit cell is shown in fig. S9 (Model 2).
- Chemical modification: Starting from Model 2 and keeping the same cell parameters, four Sr atoms have been replaced with Ca atoms, while two Sr atoms have been substituted by La atoms (Model 3).

In Model 4, the cubic unit cell of SrTiO_3 has been modified, substituting a Sr atom with a La atom. In order to simulate the structural and chemical modifications that occurred after exposing the $\text{La}_{0.2}\text{Sr}_{0.25}\text{Ca}_{0.45}\text{TiO}_3$ to H_2 atmosphere, Model 5 was built: it has the same cell parameters of Model 3 with the addition of an oxygen vacancy.

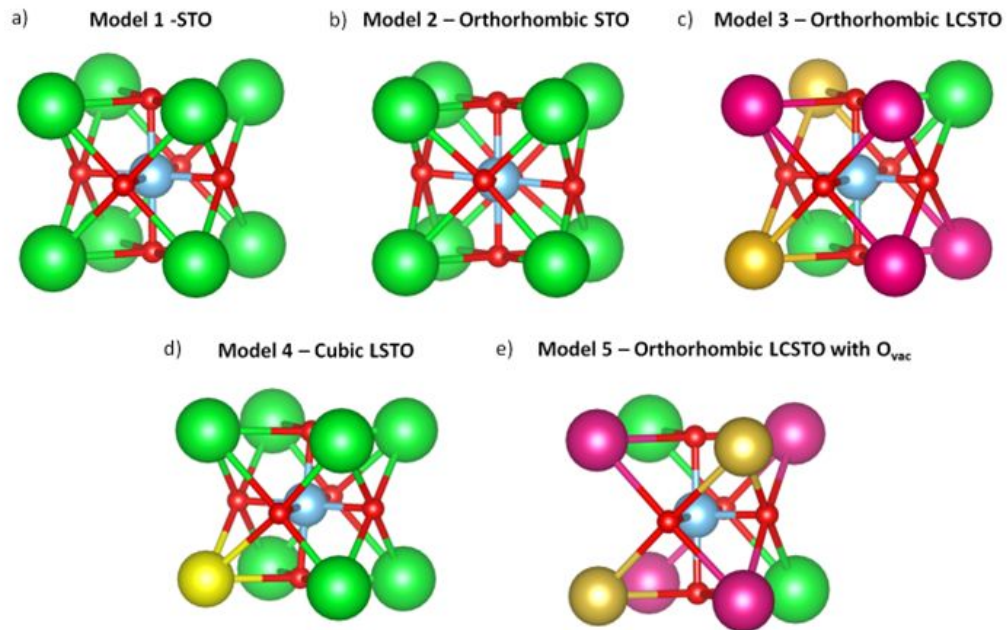


Figure S9 – a) Model 1: Cubic Unit Cell of SrTiO₃; b) Model 2: Unit cell of SrTiO₃ where cell parameters have been substituted with the ones of La_{0.2}Sr_{0.25}Ca_{0.45}TiO₃, resulting in an orthorhombic unit cell; c) Model 3: Orthorhombic Unit cell used to simulate the structure of La_{0.2}Sr_{0.25}Ca_{0.45}TiO₃. d) Model 4: Cubic unit cell of SrTiO₃ where one Sr atom has been substituted by a La atom. e) Model 5: Orthorhombic Unit cell used to simulate the structure of La_{0.2}Sr_{0.25}Ca_{0.45}TiO₃, with an oxygen vacancy. Green: Strontium; Red: Oxygen; Blue: Titanium; Yellow: Lanthanum; Purple: Calcium.

For model 3, it should be taken into account that La, Ca and Sr atoms can be arranged in different configurations in the first coordination shell of Ti. Considering as a first approximation a cubic unit cell, the number of unique permutations of the vertices is 420. When also symmetry is included, this number reduces to 17 different ways to arrange the atoms. For this reason, we performed separate calculations for the seventeen different configurations when calculating Ti⁴⁺ and Ti³⁺ L edges of LCSTO: the results are reported in Fig. S10

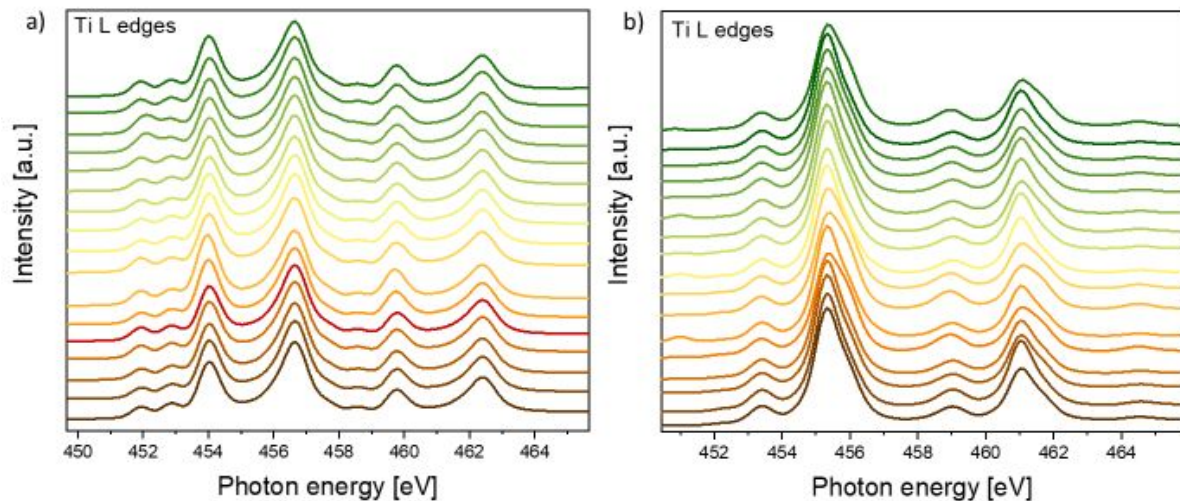


Figure S10 – Ti^{4+} (left) and Ti^{3+} (right) L edges simulations of LCSTO, considering the seventeen different configuration in which La, Ca and Sr can be arranged in model 3.

Fig. S10 shows clearly that the differences detected in the different spectra are almost negligible for the purpose of these simulations: it can be indeed observed that the position and intensity of the main features are very similar. Since the experimental spectra we show in this work are realistically a mean of all the possible configurations, we averaged these simulated spectra to obtain the calculated ones we show in the main manuscript. Clusters analogous to the one of STO have been built for model 2-5, according to their crystallographic cell parameters.

Input structure

As an example, the input structure employed for SrTiO_3 is reported hereafter. We highlighted the most important parameters that will be described in the next paragraph. The other keywords used in the input are described in detail in the FDMNES manual (<http://www.neel.cnrs.fr/fdmnes>, J. Phys. : Condens. Matter 21, 345501 (2009)).

Range

-5 .1 10 0.2 20 1 40 2 50

Energpho

Edge

L23

E_cut

-3.0

Overlap

0.19

Spinorbite

Quadrupole

TDDFT

Green

Radius

6.00

Absorber

1

Atom

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22 3 3 2 0 4 0 0 4 1 4

8 2 2 0 2 2 1 4

molecule

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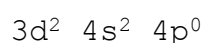
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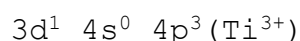
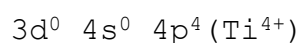
Convolution

End

The electronic configuration of the valence orbitals has been explicitly defined in the input structure, in order to be able to differentiate Ti atoms with different oxidation states. The electronic configuration in the ground state of the valence atoms for Ti⁰ is the following:



In order to simulate a Ti photoabsorber with a +4 or +3 oxidation state while keeping the whole cluster neutral, four electrons (two from the 3d and two from the 4s atomic orbitals) and three electrons (one from the 3d and two from the 4s orbitals) were moved to the 4p atomic orbitals, respectively, obtaining the following configurations:



For the simulations of SrTiO₃ and LaTiO₃ compounds, the default convolution parameters of FDMNES code have been adopted (cit manual FDMNES), i.e. using an “arctangent” shape for the broadening. For La_{0.2}Sr_{0.25}Ca_{0.45}TiO₃, a Gaussian broadening (FWHM = 0.5) has been employed to better reproduce the experimental spectrum.

Comparison between pre-edge structures of calculated STO and LCSTO NEXAFS spectra

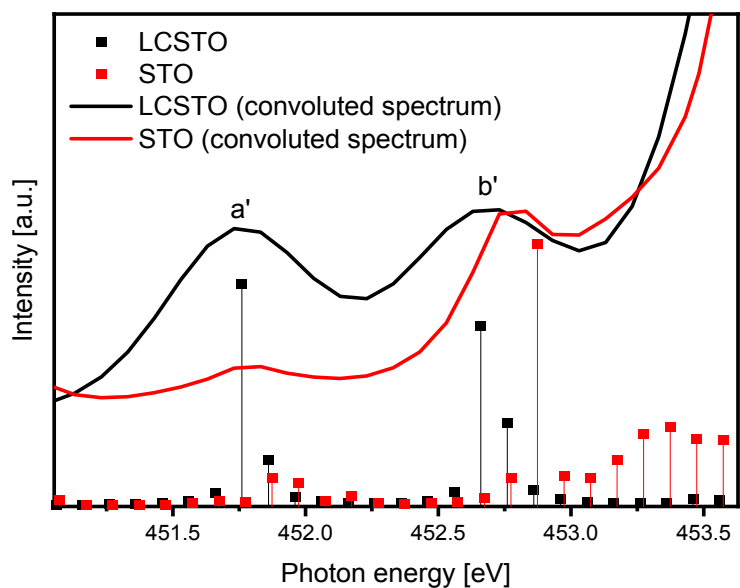


Figure S11 – Pre-edge structures of the calculated spectra of SrTiO₃ (red) and La_{0.2}Sr_{0.25}Ca_{0.45}TiO₃ (black).

Bars: non convoluted spectra; Solid lines: convoluted spectra.

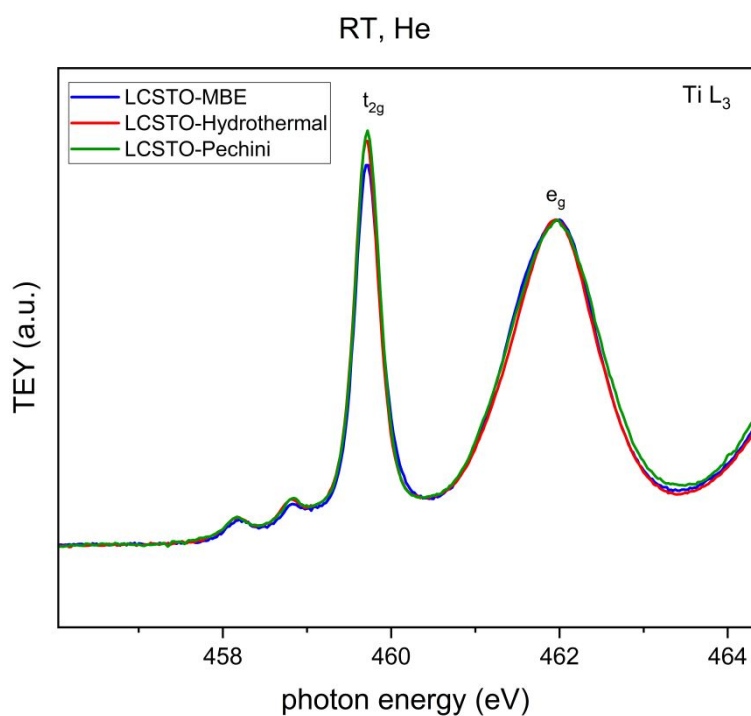


Figure S12 – Ti L₃ edge for LCSTO samples: RT in He

SrTiO₃ .cif file

For detailed information please see under Terms & Conditions.

Full information about CCDC and FIZ Karlsruhe data access policies and


```
# citation guidelines are available at http://www.ccdc.cam.ac.uk/access/V1
#
# Audit and citation data items may have been added by FIZ Karlsruhe.
# Please retain this information to preserve the provenance of
# this file and to allow appropriate attribution of the data.
#
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_symmetry_equiv_pos_site_id

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21 '-x, -z, -y'
22 '-z, -x, -y'
23 '-y, -z, -x'
24 '-x, -y, -z'
25 '-z, -y, x'
26 '-y, -x, z'
27 '-x, -z, y'
28 '-z, -x, y'
29 '-y, -z, x'
30 '-x, -y, z'
31 '-z, y, -x'
32 '-y, x, -z'
33 '-x, z, -y'
34 '-z, x, -y'
35 '-y, z, -x'
36 '-x, y, -z'
37 'z, -y, -x'
38 'y, -x, -z'
39 'x, -z, -y'
40 'z, -x, -y'
41 'y, -z, -x'
42 'x, -y, -z'
43 'z, y, x'
44 'y, x, z'
45 'x, z, y'
46 'z, x, y'
47 'y, z, x'

```
48 'x, y, z'

loop_
  _atom_site_label
  _atom_site_type_symbol
  _atom_site_fract_x
  _atom_site_fract_y
  _atom_site_fract_z
Sr1 Sr2+ 0 0 0
Ti1 Ti4+ 0.5 0.5 0.5
O1 O2- 0 0.5 0.5

#End of data_80871-ICSD
```

LaTiO₃.cif file

```
#####
#
# This file contains crystal structure data downloaded from the
# Cambridge Structural Database (CSD) hosted by the Cambridge
# Crystallographic Data Centre (CCDC) in cooperation with FIZ Karlsruhe.
#
# Please note that these data are only for research purposes or private
# use.
# For detailed information please see under Terms & Conditions.
# Full information about CCDC and FIZ Karlsruhe data access policies and
# citation guidelines are available at http://www.ccdc.cam.ac.uk/access/V1
#
# Audit and citation data items may have been added by FIZ Karlsruhe.
# Please retain this information to preserve the provenance of
# this file and to allow appropriate attribution of the data.
```

#

#####

data_98414-ICSD

_database_code_depnum_ccdc_archive 'CCDC 1657687'

loop_

_citation_id

_citation_doi

_citation_year

1 10.1103/PhysRevB.68.060401 2003

_audit_update_record

;

2018-02-27 deposited with the CCDC. 2022-02-09 downloaded from the CCDC.

;

_database_code_ICSD 98414

_chemical_name_systematic 'Lanthanum Titanate(III)'

_chemical_formula_sum 'La1 O3 Ti1'

_cell_length_a 5.6336

_cell_length_b 5.6156

_cell_length_c 7.9145

_cell_angle_alpha 90

_cell_angle_beta 90

_cell_angle_gamma 90

_cell_volume 250.38

_cell_formula_units_Z 4

_symmetry_space_group_name_H-M 'P b n m'

_symmetry_Int_Tables_number 62

_symmetry_cell_setting orthorhombic

```
loop_
_symmetry_equiv_pos_site_id
_symmetry_equiv_pos_as_xyz
1 '-x+1/2, y+1/2, z'
2 'x, y, -z+1/2'
3 'x+1/2, -y+1/2, z+1/2'
4 '-x, -y, -z'
5 'x+1/2, -y+1/2, -z'
6 '-x, -y, z+1/2'
7 '-x+1/2, y+1/2, -z+1/2'
8 'x, y, z'
```

```
loop_
_atom_site_label
_atom_site_type_symbol
_atom_site_fract_x
_atom_site_fract_y
_atom_site_fract_z
La1 La3+ 0.9916 0.0457 0.25
Ti1 Ti3+ 0.5 0 0
O1 O2- 0.0799 0.4913 0.25
O2 O2- 0.7096 0.2941 0.0417
```

```
#End of data_98414-ICSD
```

INPUT STRUCTURE FOR LCSTO

Range

```
-5 .1 10 0.2 20 1 40 2 50
```

Energpho

Edge

L23

E_cut

-3.0

Overlap

0.19

Spinorbite

Quadrupole

TDDFT

Green

Radius

6.00

Absorber

1

Atom_conf

1 1 3 3 2 0 4 0 0 4 1 4

molecule

1 1 1 90 90 90

22	0.0000000000	0.0000000000	0.0000000000
8	0.0000000000	-1.9436000000	0.0000000000
8	0.0000000000	1.9436000000	0.0000000000
8	-1.9593500000	0.0000000000	0.0000000000
8	1.9593500000	0.0000000000	0.0000000000
8	0.0000000000	0.0000000000	-2.7575350000

22	0.0000000000	-3.8872000000	0.0000000000
22	0.0000000000	3.8872000000	0.0000000000
20	-1.9593500000	-1.9436000000	2.7575350000
20	-1.9593500000	-1.9436000000	-2.7575350000
20	-1.9593500000	1.9436000000	2.7575350000
20	1.9593500000	-1.9436000000	-2.7575350000
57	-1.9593500000	1.9436000000	-2.7575350000
57	1.9593500000	-1.9436000000	2.7575350000
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38	1.9593500000	1.9436000000	2.7575350000
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22	3.9187000000	0.0000000000	0.0000000000
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8	-1.9593500000	-3.8872000000	0.0000000000
8	1.9593500000	3.8872000000	0.0000000000
8	1.9593500000	-3.8872000000	0.0000000000
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8	3.9187000000	1.9436000000	0.0000000000
8	-3.9187000000	1.9436000000	0.0000000000
8	-3.9187000000	-1.9436000000	0.0000000000
8	0.0000000000	-3.8872000000	2.7575350000
8	0.0000000000	3.8872000000	2.7575350000
8	0.0000000000	3.8872000000	-2.7575350000
8	0.0000000000	-3.8872000000	-2.7575350000
8	3.9187000000	0.0000000000	-2.7575350000
8	-3.9187000000	0.0000000000	2.7575350000
8	-3.9187000000	0.0000000000	-2.7575350000
8	3.9187000000	0.0000000000	2.7575350000
22	0.0000000000	0.0000000000	-5.5150700000
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22	3.9187000000	3.8872000000	0.0000000000
22	-3.9187000000	3.8872000000	0.0000000000
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8	0.0000000000	5.8308000000	0.0000000000
8	0.0000000000	-1.9436000000	5.5150700000

8	0.0000000000	-1.9436000000	-5.5150700000
8	0.0000000000	1.9436000000	-5.5150700000
8	0.0000000000	1.9436000000	5.5150700000
8	1.9593500000	0.0000000000	-5.5150700000
8	-1.9593500000	0.0000000000	-5.5150700000
8	1.9593500000	0.0000000000	5.5150700000
8	-1.9593500000	0.0000000000	5.5150700000
8	-5.8780500000	0.0000000000	0.0000000000
8	5.8780500000	0.0000000000	0.0000000000
8	3.9187000000	-3.8872000000	-2.7575350000
8	-3.9187000000	-3.8872000000	-2.7575350000
8	3.9187000000	-3.8872000000	2.7575350000
8	3.9187000000	3.8872000000	-2.7575350000
8	-3.9187000000	3.8872000000	-2.7575350000
8	-3.9187000000	3.8872000000	2.7575350000
8	3.9187000000	3.8872000000	2.7575350000
8	-3.9187000000	-3.8872000000	2.7575350000
38	-1.9593500000	5.8308000000	2.7575350000
38	1.9593500000	-5.8308000000	-2.7575350000
38	-1.9593500000	-5.8308000000	-2.7575350000
38	-1.9593500000	5.8308000000	-2.7575350000
38	-1.9593500000	-5.8308000000	2.7575350000
38	1.9593500000	5.8308000000	-2.7575350000
38	1.9593500000	5.8308000000	2.7575350000
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22	0.0000000000	3.8872000000	5.5150700000
22	0.0000000000	-3.8872000000	5.5150700000
22	0.0000000000	3.8872000000	-5.5150700000
22	-3.9187000000	0.0000000000	-5.5150700000
22	3.9187000000	0.0000000000	-5.5150700000
22	-3.9187000000	0.0000000000	5.5150700000
22	3.9187000000	0.0000000000	5.5150700000
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38	-5.8780500000	-1.9436000000	2.7575350000
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38	5.8780500000	-1.9436000000	-2.7575350000

38	-5.8780500000	1.9436000000	2.7575350000
38	5.8780500000	-1.9436000000	2.7575350000
38	5.8780500000	1.9436000000	-2.7575350000
38	5.8780500000	1.9436000000	2.7575350000
8	-3.9187000000	5.8308000000	0.0000000000
8	3.9187000000	5.8308000000	0.0000000000
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8	-3.9187000000	-5.8308000000	0.0000000000
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8	-1.9593500000	-3.8872000000	-5.5150700000
8	-1.9593500000	3.8872000000	-5.5150700000
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8	-1.9593500000	3.8872000000	5.5150700000
8	-1.9593500000	-3.8872000000	5.5150700000
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8	-3.9187000000	-1.9436000000	-5.5150700000
8	3.9187000000	-1.9436000000	5.5150700000
8	-3.9187000000	1.9436000000	5.5150700000
8	3.9187000000	-1.9436000000	-5.5150700000
8	3.9187000000	1.9436000000	-5.5150700000
8	3.9187000000	1.9436000000	5.5150700000
8	-3.9187000000	1.9436000000	-5.5150700000
8	-3.9187000000	-1.9436000000	5.5150700000
8	5.8780500000	-3.8872000000	0.0000000000
8	-5.8780500000	-3.8872000000	0.0000000000
8	5.8780500000	3.8872000000	0.0000000000
8	-5.8780500000	3.8872000000	0.0000000000
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22	-3.9187000000	3.8872000000	-5.5150700000
22	3.9187000000	3.8872000000	5.5150700000
22	-3.9187000000	-3.8872000000	-5.5150700000
22	3.9187000000	-3.8872000000	5.5150700000
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22	-3.9187000000	-3.8872000000	5.5150700000

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8	1.9593500000	-7.7744000000	0.0000000000
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8	0.0000000000	-5.8308000000	5.5150700000
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8	7.8374000000	0.0000000000	-2.7575350000

Convolution

Gaussian

0.5

End

- **References**

- (1) Yong, Z.; Linghu, J.; Xi, S.; Yin, X.; Leek, M. L.; Shen, L.; Timm, R.; Wee, A. T. S.; Feng, Y. P.; Pan, J. Unravelling Uniaxial Strain Effects on Electronic Correlations, Hybridization and Bonding in Transition Metal Oxides. *Acta Mater* **2019**, *164*, 618–626. <https://doi.org/10.1016/j.actamat.2018.11.017>.
- (2) De Francesco, R.; Stener, M.; Fronzoni, G. Computational Investigation of the L_{2,3}-Edge Spectra of Bulk and (110) Surface of Rutile TiO₂. *Surf Sci* **2011**, *605* (5–6), 500–506. <https://doi.org/10.1016/j.susc.2010.12.006>.
- (3) De Groot, F. M. F.; Fuggle, J. C.; Thole, B. T.; Sawatzky, G. A. *L q 3 X-Ray-Absorption Edges of d Compounds: K⁺, Ca²⁺, Sc³⁺, and Ti³⁺ in Oh (Octahedral) Symmetry*; 1990; Vol. 8.