



The effect of artificial weathering on PP coextruded tape and laminate



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ABSTRACT

The aim of this work was to analyse the influence of artificial weathering on polypropylene (PP) self-reinforced composite both as fabric constituent (tape) and as laminate. Particular attention has been focused on the interaction between radiation and material microstructural characteristics, especially for the selective action that the former causes in PP amorphous regions. The evaluation of carbonyl index performed by Fourier Transform Infrared (FT-IR) spectroscopy has highlighted that tapes are more prone to degradation since their drawn structure induces internal stress. Differential scanning calorimetry (DSC) measurements have denoted a shift towards lower temperature of PP melting peak. While crystallinity determination performed by DSC and FT-IR spectroscopy has underlined an increasing trend for laminate over exposure time due to the higher amount of amorphous phase, Raman spectroscopy has revealed that photo-degradation induces a rise of the isomeric defect fraction, limiting chemi-crystallisation both for tapes and laminates.

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1. Introduction

Thermoplastic composite investigation is one of the most explored research fields since manufacturers are interested to gain materials with superior characteristics in terms of mechanical resistance and reduced weight. In order to introduce innovative products on the market, many studies have been conducted on polypropylene based Self-Reinforced Composites (SRCs) which seems to be a promising material class for several industrial applications [1,2].

Several manufacturing processes have been reviewed in literature [1,2], but one of the most attractive production technologies is tape coextrusion. Coextruded tapes, composed of high stretched polypropylene (PP) homopolymer core and lower melting temperature polypropylene/polyethylene copolymer (coPP/PE) skin, have been widely studied [3–5] in order to evaluate how drawing process affects material characteristics. Particular attention has been paid to the influence of consolidation parameters on laminate mechanical properties [6,7] and impact resistance [8].

Since SRCs can be designed for outdoor applications, the influence of atmospheric agents on final performance has to be considered. Many factors such as heat and oxygen [9–11], water absorption [12] and low temperature [13] play a key-role in PP degradation mechanisms, but the most aggressive agent is the ultraviolet radiation.

The effects of photo-oxidation on polyolefin (both PP homopolymer and coPP/PE) have been already studied in literature. A rapid decrease of tensile strength can be observed in PP injection moulded samples at different exposure time [14,15] and in PP filament yarns subjected to natural and artificial weathered conditions [16]. Similarly, the Young's modulus of PP extruded plates decreases with exposure time as a consequence of crack formation on material surface [17].

Different techniques have been used to study degradation mechanisms through the detection of low molecular by-products formed during reactions. Typical functional groups, such as hydroxyl (OH), hydroperoxide (OOH) [18] and in particular carbonyl (C=O) [19,20], have been identified by FT-IR. On the other hand, Raman spectroscopy is more suitable for vinyl groups (—CH=CH₂) determination [21,22] and for the creation of spectral maps on microscopic scale in order to define species distribution on sample surface [23].

Chain scission and crosslinking rates have been determined for thermoplastics (polypropylene, polyethylene and copolymer) by means of gel permeation chromatography [24]. It has been noted that reaction rates are very effective near material surface, whereas they significantly decrease through the bars due to scarce oxygen concentration (oxygen diffusion limited reaction). The same outcomes have been found elsewhere [25], where comprehensive information on the modifications induced by outdoor aging on coPP/PE are provided.

The effect caused by natural aging on isotactic polypropylene (i-PP) has been analysed by means of Differential Scanning

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Calorimetry (DSC) and Thermogravimetry (TG) in order to evaluate crystallisation heat variations [16]. Material characteristics, such as crystallinity and spherulite size, influence photo-oxidation rate [26]: in particular higher crystallinity fraction leads to slower degradation rate. During artificial weathering a crystallinity increase can be observed due to chemi-crystallisation, a mechanism that induces molecular reorganisation in PP amorphous phase [27], with a subsequent melting enthalpy rise. Furthermore, for samples recrystallised under non-isothermal conditions, longer exposure time results in (i) crystallinity fraction reduction, (ii) fusion peak duplication and (iii) melting temperature decrement, due to impurities, reorganisation during heating and low molecular weight formation (chain scission) respectively [28].

Some authors [29,30] have studied the synergistic effect of stress and UV irradiation on PP. The results have shown that both chain scission and crosslinking occur on material surface and their concentration increases accordingly to tensile stress [29].

Slow strain rate tensile test performed on notched samples have revealed that cracking and chemical degradation are enhanced by stress concentration in the notch-surrounding area. Furthermore, sample mechanical properties are reduced if UV irradiation has been applied during tensile test [30].

In the field of SRCs a preliminary study has been conducted to determine qualitatively the difference in matrix and reinforcement after weathering by means of Raman spectroscopy [31].

This work aims to evaluate UV effect on coextruded tape and laminate for a possible outdoor application. SRC mechanical properties as a function of exposure time have been already investigated in a previous work [31], where a significant decrease of mechanical properties and an evident embrittlement after 250 h have been highlighted.

In this study particular attention has been focused on the interaction between radiation and the highly drawn microstructure of this material class. After a visual inspection, DSC, FT-IR and Raman spectroscopy have been applied to quantify the degradation, measuring melting temperature, carbonyl index and crystallinity.

2. Materials and methods

A commercial plain weave fabric has been selected for the present work. The fabric has an overall weight of 107 g/m² and it is composed of coextruded PP tapes 2.5 mm wide with a linear density of 1300 dTex. The fabric has the same number of tapes arranged in warp and weft direction, with C:H:C tape cross-section configuration, where the skin C is coPP/PE + PE blend and H is the oriented PP homopolymer. Ten fabric layers have been positioned between 300 × 300 mm hot plates and samples have been consolidated at 140 °C and 2.2 MPa, as suggested by manufacturer's specifications.

2.1. Artificial weathering

A Q-Sun Xenon test Chamber Model Xe-1 manufactured by Q-LAB, equipped with a xenon-arc lamp, which emits radiation from ultraviolet to infrared, has been used for accelerated aging. A daylight filter has been applied in order to produce a spectral power distribution (SPD) of 678.8 Wm⁻² in the range 300–800 nm equivalent to noon summer sunlight. An irradiance sensor and an insulated black panel temperature sensor, both embedded in the chamber, control the wavelength in the range 340 ± 10 nm and the temperature, respectively. The exposure parameters used during artificial weathering test have been chosen in accordance with ISO 4892-part 2.

2.2. Differential Scanning Calorimetry (DSC)

Differential Scanning Calorimetry (DSC) has been performed with a Netzsch DSC 200 F3 Maia[®] differential scanning calorimeter. 5 mg samples, obtained randomly both from tape and consolidated laminates, placed within closed aluminium crucibles, have been heated from room temperature to 200 °C with a heating rate of 10 °C min⁻¹. Three samples have been measured for each test performed. DSC traces have been analysed with Netzsch Proteus – Thermal Analysis – Version 6.1.0.

2.3. Raman spectroscopy

An InVia Raman microscope (Renishaw plc., Wotton-Under-Edge, UK) has been used to acquire all Raman spectra. The laser (785 nm diode laser delivering 250 mW to the sample) has been focused by a microscope objective (10× N.A. 0.25) onto the sample. Six spectra have been collected randomly around the samples with an acquisition time of 30 s.

Spectra have been analysed using hyperSpec [32], a package of the R software for statistical analysis [33] and Fityk 0.9.8 [34] fitting the vibrational bands with Voigt function. The fraction of the crystalline phase (X_c^{Raman}) and the isomeric defect phase (X_b^{Raman}) have been evaluated using the method described by Nielsen [35].

2.4. FT-IR spectroscopy

Three FT-IR spectra of each sample have been collected between 400 and 4000 cm⁻¹ using a Thermo Nicolet Nexus 470 FTIR with Avatar Diffuse Reflectance accessory. The nominal resolution of the spectrometer has been 2 cm⁻¹, averaging 120 scans. Samples have been prepared using a Silicon carbide sampling kit.

As for Raman spectroscopy, spectra have been analysed using hyperSpec [32], a packages of the R software for statistical analysis [33] and Fityk 0.9.8 [34] fitting the vibrational bands with Voigt function.

The crystalline phase fraction ($X_c^{\text{FT-IR}}$) has been evaluated using the method described by Karacan [36]. Following the work of Rabello [26,27] and Aslanzadeh [18], the Carbonyl Index ($CI^{\text{FT-IR}}$) has been evaluated as the ratio between the area of the absorbance band in the range of 1650–1850 cm⁻¹ and the area of the absorbance band in the region of 2694–2750 cm⁻¹.

3. Results and discussions

Fig. 1 shows laminate and tape exterior morphology before irradiation and after 500 h of exposure.

As it can be noticed, tape and laminate show negligible defects, probably caused by manufacturing process; conversely, laminate surface exposed to radiation is characterized by the presence of cracks, which are more evident and widely distributed than on tape.

This fact can be explained by chemi-crystallisation, a mechanism that involves molecular reorganisation only in the amorphous regions of the polymer (higher for laminate due to consolidation process), that causes spontaneous crack formation due to surface layer contraction [27].

3.1. FT-IR and Raman spectroscopy

Fig. 2 depicts typical FT-IR spectra of laminate and tape after different irradiation periods.

After irradiation the main spectral changes occur in the region between 1650 and 1850 cm⁻¹, where the characteristic peaks associated to carbonyl group vibrations appear and increase for longer exposure times. After 250 h, another broad band between 3250

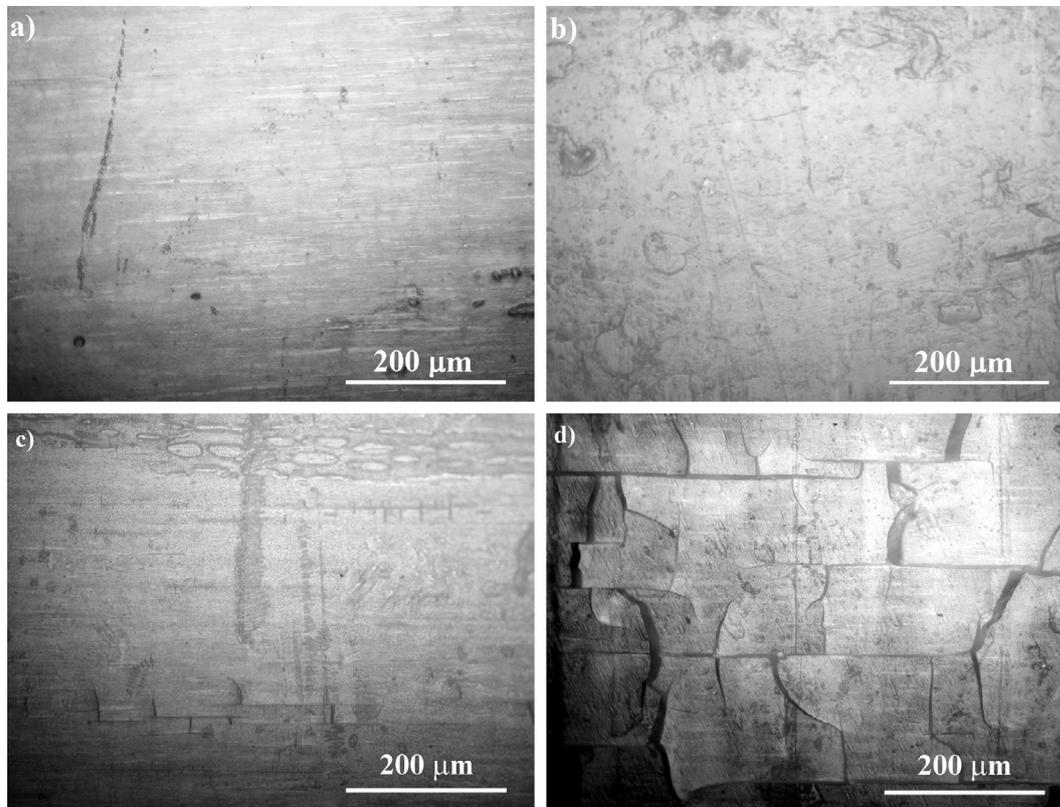


Fig. 1. Surface morphology of tape and laminate: before irradiation (a) and (b), after 500 h of exposure (c) and (d).

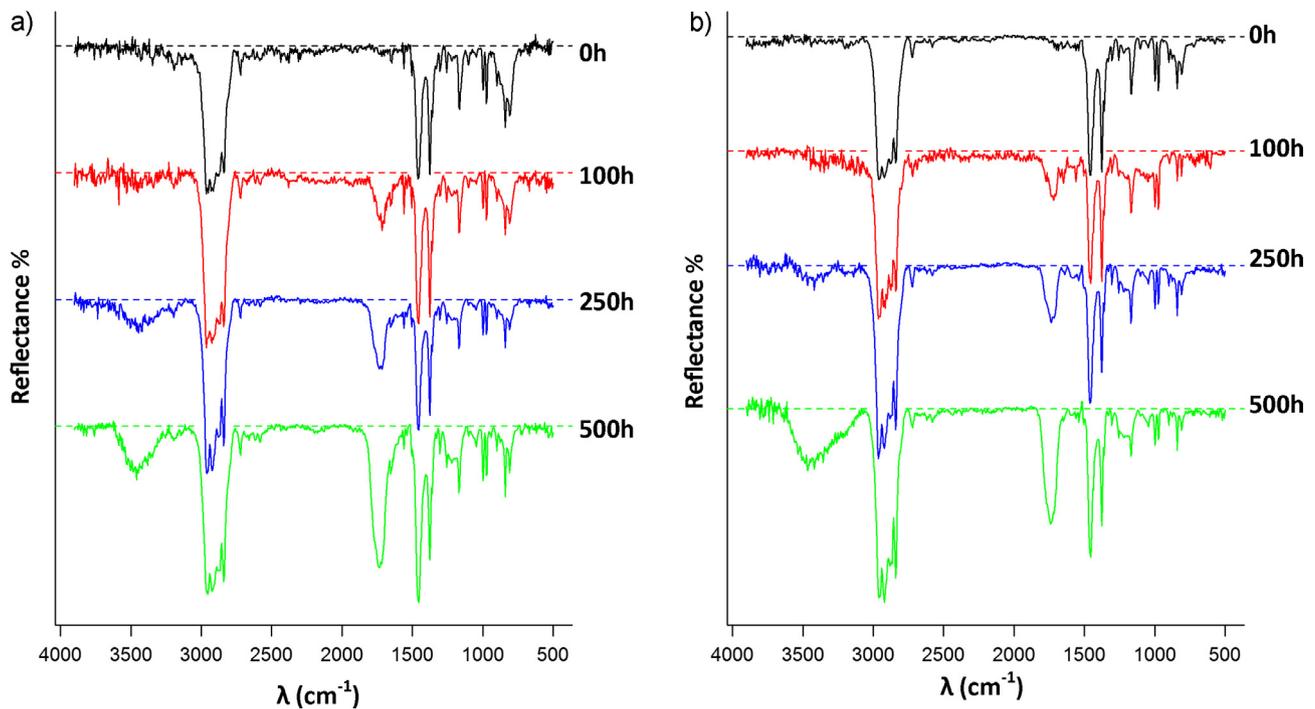


Fig. 2. FT-IR spectra of tape (a) and laminate (b) at increasing exposure time: no exposure (black line), 100 h (red line), 250 h (blue line), 500 h (green line). (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

and 3600 cm^{-1} , corresponding to hydroxyl and/or hydroperoxide group vibrations [18], appears and its intensity increases up to 500 h.

As expected, Raman spectra do not show any changes in the region between 1650 and 1850 cm^{-1} due to Raman spectroscopy selection rules [23]. But, as reported in previous work [37], Raman

spectroscopy is able to highlight the presence of PE vibrational bands, both in tape and laminate spectra.

Carbonyl Index (CI^{FT-IR}) has been calculated both for laminate and tape, following the work of Rabello [26,27] and Aslanzadeh [18], in order to quantify carbonyl group formation over time.

As expected, both laminate and tape CI^{FT-IR} increases with exposure time (Fig. 3). For tape, CI^{FT-IR} values are higher already after 100 h and 250 h of exposure, probably because of a synergistic effect of chemical degradation and tape internal stress induced by drawing process [30]. Conversely to tapes, laminates are less susceptible to chemical degradation because consolidation partially removes those internal stresses.

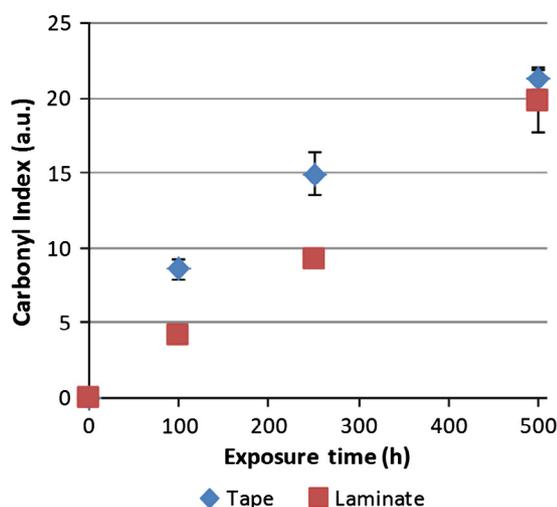


Fig. 3. CI^{FT-IR} as a function of exposure time. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

3.2. Differential Scanning Calorimetry (DSC)

Fig. 4 summarises laminate and tape DSC traces at increasing exposure time. As reported in previous work [37], melting thermograms of both tape (Fig. 4a, black line) and laminate (Fig. 4b, black line) samples showed distinct melting peaks for PE (107 °C) and PP (between 140 and 170 °C); after data fitting, three melting peaks for PP during first heating have been identified.

Tape main melting peak temperature (166 °C) appears to be stable up to 100 h of irradiation. After 250 h and 500 h (Fig. 4a, blue and green curves) main melting peak shifts toward lower value, about 158 °C and 153 °C respectively, while a small melting peak appears at 166 °C, positioned at the same temperature of the unexposed sample main peak.

Laminates show the same behaviour of tapes, even if melting temperature shift is lower, as highlighted in Fig. 5. Samples after 100 h (Fig. 4b, red curve) show a melting peak at 167 °C (3 °C lower respect to unexposed samples). Specimens exposed for 250 h (Fig. 4b, blue curve) show a double feature (main melting peak at 160 °C and a secondary one at 165 °C); after 500 h (Fig. 4b, green curve) the main peak (160 °C) becomes more prominent while the secondary (still clearly visible at 165 °C) becomes less pronounced.

As underlined above and as noticeable in Fig. 5, PP melting peak shifts towards lower values at increasing exposure times. Chain scission, due to photo-degradation, occurs mainly in the amorphous region, but also at the lamellar fold surface, causing an increase of crystal surface free energy and a melting temperature decrease [26,38]. As above mentioned, melting temperature decrease is more pronounced in exposed tapes than in laminates, because chemical degradation is enhanced by internal stress [30].

The presence of a double peak in exposed sample thermographs is caused by the formation of small and defective molecules, like carbonyl and hydroperoxides groups, during the photo-degradation process. These small molecules contribute to the

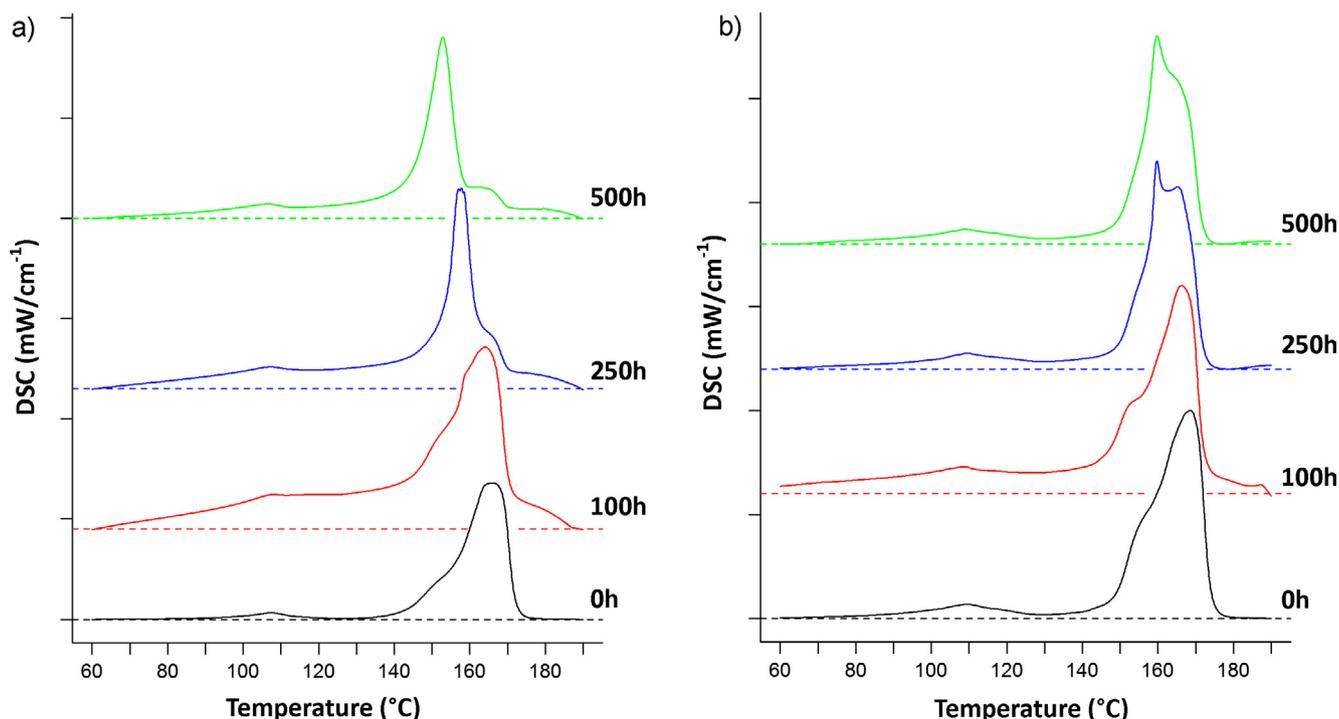


Fig. 4. DSC traces of tape (a) and laminate (b) at increasing exposure time: no exposure (black line), 100 h (red line), 250 h (blue line), 500 h (green line). (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

recrystallisation process where imperfect and heterogeneous crystals are formed [38].

It is worth noticing that PE melting temperature (107 °C) remains constant over aging time for all samples.

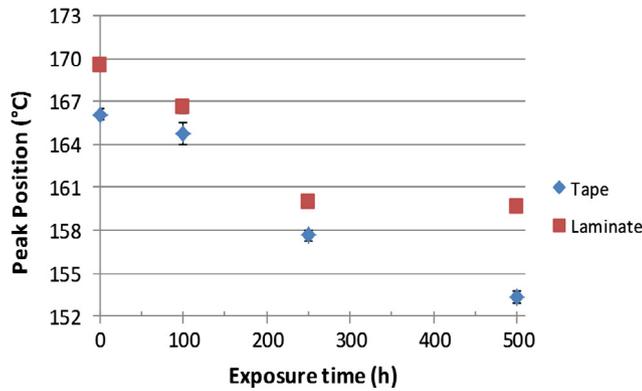


Fig. 5. Main melting peak position as a function of exposure time. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

3.3. Crystallinity evaluation

Crystallinity calculated using Raman, DSC and FT-IR as a function of exposure time is shown in Figs. 6 and 7. Those values are reasonably comparable, since it has been already reported that crystallinity determination depends on the technique used [35,39]. It is noteworthy to mention that crystallinity values calculated using Raman spectroscopy are more affected by the material lack of homogeneity [37] because spectra are collected by means of a microscope objective, that analyses a smaller area than DSC and FT-IR. In addition, with Raman spectroscopy it is possible to estimate both crystalline phase (X_c^{Raman}) and isomeric defect phase (X_b^{Raman}), consisting of helical chains (Fig. 7).

The high PP crystallinity value for SRC has been already observed in a previous work [37], and its explanation is related to the high draw ratio applied during coextrusion of PP (core) and coPP/PE (skin), that causes a stretching-induced crystallisation [2].

The lower crystallinity value found for laminates can be ascribed to the portion of material that melts and becomes amorphous during the consolidation process conducted at 140 °C. At increasing exposure time, this amorphous part can rearrange its structure and increase the overall crystallinity [26,27].

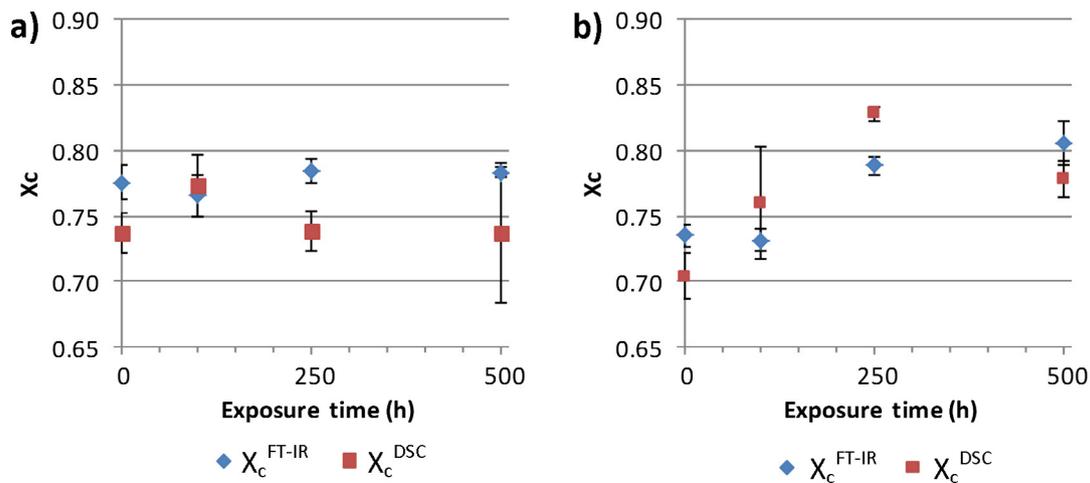


Fig. 6. Crystallinity fraction as a function of exposure time: comparison between FT-IR (X_c^{FT-IR}) and DSC (X_c^{DSC}) for tapes (a) and for laminates (b). (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

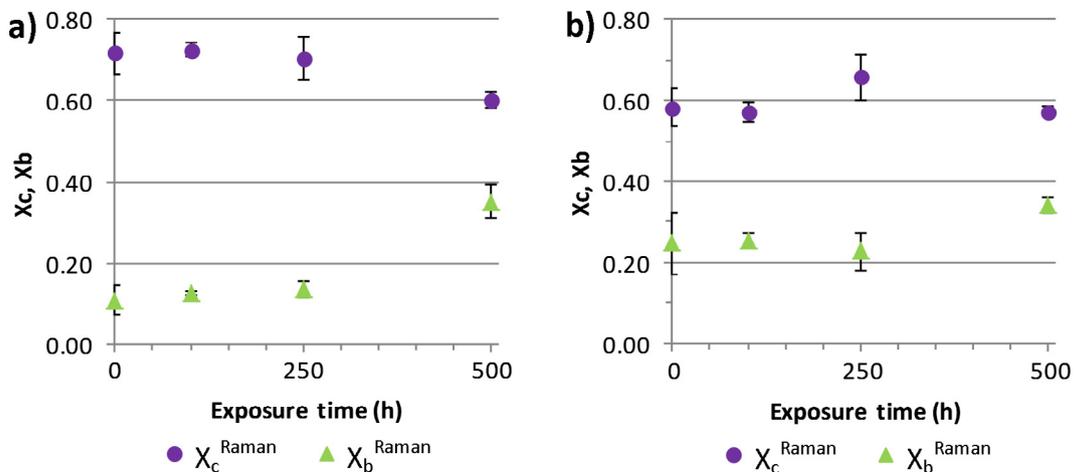


Fig. 7. Crystallinity fraction (X_c^{Raman}) and isomeric defect phase (X_b^{Raman}) as a function of exposure time for tapes (a) and laminates (b). (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

From FT-IR and DSC data, tape crystallinity (Fig. 6a) seems to be not affected by the exposure time, while laminate crystallinity increases after 100 h of exposure (Fig. 6b) reaching a constant value at 250 h.

Tape X_c^{Raman} decreases after 500 h of exposure, while the amount of isomeric defect phase increases (Fig. 7a). As above mentioned, laminate crystallinity before exposure is lower than tape, but the isomeric defect fraction (0.25 ± 0.08) is higher for laminate than for tape (0.11 ± 0.04). Hence the consolidation at 140°C influences the amount of both crystalline and isomeric defect phase. During exposure laminate X_c^{Raman} slightly increases until 250 h and then decreases; on the contrary the isomeric defect phase fraction (X_b^{Raman}) remains constant up to 250 h and then increases at 500 h.

After prolonged exposure time the generation of impurity groups like carbonyls and hydroperoxides reduces the molecular regularity [27], limits the chemi-crystallisation and, as highlighted by Raman results, increases the isomeric defect fraction.

4. Conclusions

In this work the effect of artificial weathering on polypropylene tapes and laminates has been evaluated using differential scanning calorimetry, Fourier Transform Infrared spectroscopy and Raman spectroscopy. Samples morphological analysis after 500 h of exposure has highlighted the presence of surface cracks, more evident for laminate due to chemi-crystallisation of amorphous PP regions. FT-IR spectroscopy has revealed the presence of low molecular weight by-products characterized by carbonyl or hydroxyl/hydroperoxide groups. In particular carbonyl index is higher for tapes than for laminates, especially after 100 h and 250 h of exposure, because of internal stress induced by drawing process.

DSC measurements have verified that PP melting peak shifts to lower values at increasing exposure time whereas PE melting temperature remains constant.

DSC and FT-IR results have demonstrated that while tape crystallinity is not affected by the exposure time, the greater amorphous phase present in laminates is able to rearrange its structure causing a rise in overall crystallinity. Furthermore, Raman spectroscopy has revealed that after prolonged exposure time, photo-degradation induces a rise of the isomeric defect fraction, limiting chemi-crystallisation both for tapes and laminates.

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References

- [1] Kmetty Á, Bárányi T, Karger-Kocsis J. Self-reinforced polymeric materials: a review. *Prog Polym Sci* 2010;35:1288–310. <http://dx.doi.org/10.1016/j.progpolymsci.2010.07.002>.
- [2] Karger-Kocsis J, Bárányi T. Single-polymer composites (SPCs): status and future trends. *Compos Sci Technol* 2014;92:77–94. <http://dx.doi.org/10.1016/j.compscitech.2013.12.006>.
- [3] Alcock B, Cabrera NO, Barkoula NM, Peijs T. The effect of processing conditions on the mechanical properties and thermal stability of highly oriented PP tapes. *Eur Polym J* 2009;45:2878–94. <http://dx.doi.org/10.1016/j.eurpolymj.2009.06.025>.
- [4] Alcock B, Cabrera NO, Barkoula N-M, Loos J, Peijs T. Interfacial properties of highly oriented coextruded polypropylene tapes for the creation of recyclable all-polypropylene composites. *J Appl Polym Sci* 2007;104:118–29. <http://dx.doi.org/10.1002/app.24588>.
- [5] Izer A, Bárányi T, Varga J. Development of woven fabric reinforced all-polypropylene composites with beta nucleated homo- and copolymer matrices. *Compos Sci Technol* 2009;69:2185–92. <http://dx.doi.org/10.1016/j.compscitech.2009.06.002>.
- [6] Alcock B, Cabrera NO, Barkoula N-M, Loos J, Peijs T. The mechanical properties of unidirectional all-polypropylene composites. *Compos Part Appl Sci Manuf* 2006;37:716–26. <http://dx.doi.org/10.1016/j.compositesa.2005.07.002>.
- [7] Alcock B, Cabrera NO, Barkoula N-M, Spoelstra AB, Loos J, Peijs T. The mechanical properties of woven tape all-polypropylene composites. *Compos Part Appl Sci Manuf* 2007;38:147–61. <http://dx.doi.org/10.1016/j.compositesa.2006.01.003>.
- [8] Alcock B, Cabrera NO, Barkoula N-M, Wang Z, Peijs T. The effect of temperature and strain rate on the impact performance of recyclable all-polypropylene composites. *Compos Part B Eng* 2008;39:537–47. <http://dx.doi.org/10.1016/j.compositesb.2007.03.003>.
- [9] Richaud E, Farcas F, Fayolle B, Audouin L, Verdu J. Hydroperoxide build-up in the thermal oxidation of polypropylene – a kinetic study. *Polym Degrad Stab* 2007;92:118–24. <http://dx.doi.org/10.1016/j.polymdegradstab.2006.08.010>.
- [10] Celina MC. Review of polymer oxidation and its relationship with materials performance and lifetime prediction. *Polym Degrad Stab* 2013;98:2419–29. <http://dx.doi.org/10.1016/j.polymdegradstab.2013.06.024>.
- [11] Fernando SS, Christensen PA, Egerton TA, White JR. Humidity dependence of carbon dioxide generation during photodegradation of biaxially oriented polypropylene in oxygen. *Polym Degrad Stab* 2009;94:83–9. <http://dx.doi.org/10.1016/j.polymdegradstab.2008.10.007>.
- [12] Deng H, Reynolds CT, Cabrera NO, Barkoula N-M, Alcock B, Peijs T. The water absorption behaviour of all-polypropylene composites and its effect on mechanical properties. *Compos Part B Eng* 2010;41:268–75. <http://dx.doi.org/10.1016/j.compositesb.2010.02.007>.
- [13] Alcock B, Cabrera NO, Barkoula N-M, Peijs T. Low velocity impact performance of recyclable all-polypropylene composites. *Compos Sci Technol* 2006;66:1724–37. <http://dx.doi.org/10.1016/j.compscitech.2005.11.010>.
- [14] Rosa DS, Angelini JMG, Agnelli JAM, Mei LHI. The use of optical microscopy to follow the degradation of isotactic polypropylene (iPP) subjected to natural and accelerated ageing. *Polym Test* 2005;24:1022–6. <http://dx.doi.org/10.1016/j.polymertesting.2005.07.009>.
- [15] Turton TJ, White JR. Observation of different photo-degradation behaviour in two similar polypropylenes. *J Mater Sci* 2001;36:4617–24.
- [16] Gallo R, Severini F. Course of the changes in thick and thin isotactic polypropylene samples subjected to natural aging. *Polym Degrad Stab* 2013;98:1144–9. <http://dx.doi.org/10.1016/j.polymdegradstab.2013.03.017>.
- [17] Yakimets I, Lai D, Guigon M. Effect of photo-oxidation cracks on behaviour of thick polypropylene samples. *Polym Degrad Stab* 2004;86:59–67. <http://dx.doi.org/10.1016/j.polymdegradstab.2004.01.013>.
- [18] Aslanzadeh S, Kish MH. Photo-oxidation of polypropylene fibers exposed to short wavelength UV radiations. *Fibers Polym* 2010;11:710–8.
- [19] Fernando SS, Christensen PA, Egerton TA, White JR. Carbon dioxide evolution and carbonyl group development during photodegradation of polyethylene and polypropylene. *Polym Degrad Stab* 2007;92:2163–72. <http://dx.doi.org/10.1016/j.polymdegradstab.2007.01.032>.
- [20] Gallo R, Brambilla L, Castiglioni C, Ipsale S, Severini F, Quasso F, et al. Environmental degradation of a novel ethylene-propylene copolymer in thick sheets. *Eur Polym J* 2005;41:359–66. <http://dx.doi.org/10.1016/j.eurpolymj.2004.09.014>.
- [21] Arruebarrena de Báez M, Hendra PJ, Judkins M. The Raman spectra of oriented isotactic polypropylene. *Spectrochim Acta A Mol Biomol Spectrosc* 1995;51:2117–24. [http://dx.doi.org/10.1016/0584-8539\(95\)01512-1](http://dx.doi.org/10.1016/0584-8539(95)01512-1).
- [22] Gen DE, Prokhorov KA, Nikolaeva GY, Sagitova EA, Pashinin PP, Shklyaruk BF, et al. Raman spectra of various polymorphs of isotactic polypropylene. *Laser Phys* 2011;21:125–9. <http://dx.doi.org/10.1134/S1054660X11020010>.
- [23] Blakey I, George GA. Raman spectral mapping of photo-oxidised polypropylene. *Polym Degrad Stab* 2000;70:269–75.
- [24] White JR, Shyichuk AV. Macromolecular scission and crosslinking rate changes during polyolefin photo-oxidation. *Polym Degrad Stab* 2007;92:1161–8. <http://dx.doi.org/10.1016/j.polymdegradstab.2007.04.011>.
- [25] Shyichuk AV, White JR, Craig IH, Syrotynska ID. Comparison of UV-degradation depth-profiles in polyethylene, polypropylene and an ethylene-propylene copolymer. *Polym Degrad Stab* 2005;88:415–9. <http://dx.doi.org/10.1016/j.polymdegradstab.2004.12.006>.
- [26] Rabello MS, White JR. The role of physical structure and morphology in the photodegradation behaviour of polypropylene. *Polym Degrad Stab* 1997;56:55–73.
- [27] Rabello MS, White JR. Crystallization and melting behaviour of photodegraded polypropylene—I. Chemi-crystallization. *Polymer* 1997;38:6379–87.
- [28] Rabello MS, White JR. Crystallization and melting behaviour of photodegraded polypropylene—II. Re-crystallization of degraded molecules. *Polymer* 1997;38:6389–99.
- [29] Shyichuk AV, Stavychna DY, White JR. Effect of tensile stress on chain scission and crosslinking during photo-oxidation of polypropylene. *Polym Degrad Stab* 2001;72:279–85.
- [30] Kelly CT, White JR. Photo-degradation of polyethylene and polypropylene at slow strain-rate. *Polym Degrad Stab* 1997;56:367–83.
- [31] Ferluga A, Caniato M, Sbaizero O. The influence of consolidation and artificial weathering on all-PP composite behavior. *J Appl Polym Sci* 2015;132:41283. <http://dx.doi.org/10.1002/app.41283>.
- [32] Beleites C, Sergio V. hyperSpec: a package to handle hyperspectral data sets in R; n.d.
- [33] R Core Team. R: A language and environment for statistical computing. R Foundation for Statistical Computing, Vienna; 2013.

- [34] Wojdyr M. Fityk: a general-purpose peak fitting program. *J Appl Crystallogr* 2010;43:1126–8. <http://dx.doi.org/10.1107/S0021889810030499>.
- [35] Nielsen AS, Batchelder DN, Pyrz R. Estimation of crystallinity of isotactic polypropylene using Raman spectroscopy. *Polymer* 2002;43:2671–6.
- [36] Karacan I, Benli H. The influence of annealing treatment on the molecular structure and the mechanical properties of isotactic polypropylene fibers. *J Appl Polym Sci* 2011;122:3322–38. <http://dx.doi.org/10.1002/app.34440>.
- [37] Marsich L, Ferluga A, Venturini N, Caniato M, Sbaizero O, Schmid C. The morphological properties of PP coextruded tape fabrics. *Polym Eng Sci*. 2016; n/a–a. <http://dx.doi.org/10.1002/pen.24299>.
- [38] Fechine GJM, Demarquette NR. Cracking formation on the surface of extruded photodegraded polypropylene plates. *Polym Eng Sci*. 2008;48:365–72. <http://dx.doi.org/10.1002/pen.20958>.
- [39] Lin W, Cossar M, Dang V, Teh J. The application of Raman spectroscopy to three-phase characterization of polyethylene crystallinity. *Polym Test* 2007;26:814–21. <http://dx.doi.org/10.1016/j.polymertesting.2007.05.004>.